

Electronic Structure of Perovskites: Lessons from Hybrid Functionals

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① Intro: Perovskites

② Computational Modeling

The Many Body Hamiltonian

HF & DFT

Hybrid Functionals

③ Results: Hybrid Functionals applied to Perovskites

The 3d LaMO_3 perovskite dataset

4d and 5d perovskites (RTcO_3 & BaIrO_3)

Metal-to-insulator (MIT) transition in LaMnO_3

MIT and polarons in BaBiO_3

Multiferroicity in PbNiO_3

Crystal Structure Prediction: SrPbO_3

Once Upon a Time in the Ural Mountains

In 1839, the Prussian mineralogist Gustav Rose discovered a new mineral, CaTiO_3 , which he named **perovskite**, in honor of the Russian mineralogist Count Lev A. Perovski.



Gustav Rose



CaTiO_3



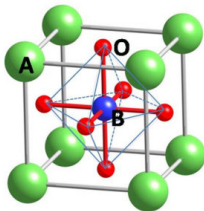
Lev A. Perovski

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ABO_3



Lev A. Perovski

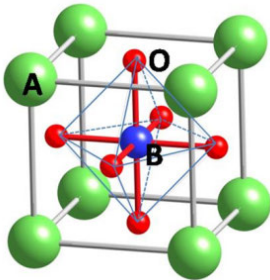
Basics: Chemistry

- $A^{2+}B^{4+}O_3^{2-}$: high chemical flexibility

A-site ion: large cation

Alkaline earth metal $\rightarrow$ Ca, Sr, Ba

Rare earth element $\rightarrow$ La, Sc, Y



The periodic table shows elements arranged by atomic number. The highlighted elements are:

- Hydrogen (H)
- Lithium (Li)
- Beryllium (Be)
- Sodium (Na)
- Magnesium (Mg)
- Scandium (Sc)
- Yttrium (Y)
- Lanthanum (La)
- Cerium (Ce)

*Lanthanide series

Scandium 17	Titanium 22	Vanadium 23	Chromium 24	Manganese 25	Iron 26	Cobalt 27	Nickel 28	Copper 29	Zinc 30	Gallium 31	Germanium 32	Arsenic 33	Selenium 34	Bromine 35	Krypton 36
La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb		
72.01	140.12	140.91	144.24	150.01	55.845	58.933	58.693	63.546	65.38	69.723	72.630	74.922	78.96	79.904	83.80

* * Actinide series

89	90	91	92	93	94	95	96	97	98	99	100	101	102
Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No
227	232	231	238	237	244	243	247	247	251	252	257	288	289

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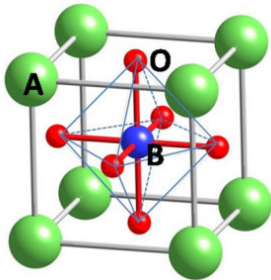
Alkaline earth metal $\rightarrow$ Ca, Sr, Ba

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B-site ion: transition metal (TM) elements

$3d \rightarrow \text{Ti, V, Mn, Fe, Co, Ni, Cu}$

$4d \rightarrow \text{Nb, Tc, Ru, Rh}$

$$5d \rightarrow \text{Os, Ir}$$
^a Lanthanide series.

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La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb		
138.91	140.12	140.91	144.24	144.91	150.36	151.96	157.25	158.93	167.30	168.93	173.05	174.97	177.05	186.90	188.91

* = Actinide series

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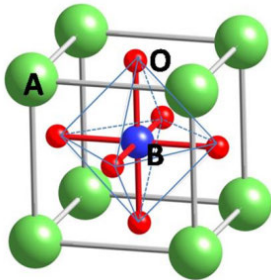
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C-site ion: Oxygen



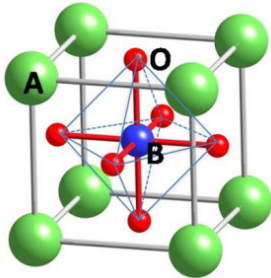
* Lanthanide series

57	58	59	60	61	62	63	64	65	66	67	68	69	70
La	Ce	Pr	Nd	Pm	Sm	Eu	Gd	Tb	Dy	Ho	Er	Tm	Yb
138.91	140.12	140.91	144.24	144.91	150.36	151.96	157.25	158.93	162.50	164.93	167.26	168.93	173.05

** Actinide series

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Ac	Th	Pa	U	Np	Pu	Am	Cm	Bk	Cf	Es	Fm	Md	No
227.03	232.04	231.04	238.03	237.05	244.06	243.06	247.07	247.07	251.08	252.08	257.10	258.10	259.10

Basics: Structure



- Goldschmidt tolerance factor t

$$t = (R_A + R_O) / \sqrt{2}(R_B + R_O)$$

$t=1$

Cubic ($R_A = R_B$)

SrTiO3

$t > 1$

Hexagonal ($R_A > R_B$)

face-sharing of octahedra

$t = 0.71 - 0.9$

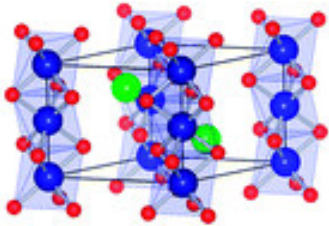
Orthorhombic ($R_A < R_B$)

Rhombohedral

$t > 0.71$

Different structures ($R_A \sim R_B$)

Basics: Structure



Fuks *et al.*, J. Mater. Chem. A 1 14320 (2013)

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(Ba,Sr)(Co,Fe)O₃, BaNiO₃

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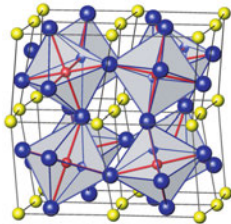
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Pavarini *et al.*, New J. Phys. 7, 188 (2005)

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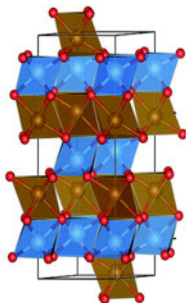
$t > 1$ Hexagonal ($R_A > R_B$)
face-sharing of octahedra

$t = 0.71 - 0.9$ Orthorhombic ($R_A < R_B$)
Rhombohedral

GdFeO₃

$t > 0.71$ Different structures ($R_A \sim R_B$)

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Riberio *et al.*, RSC Adv., 4, 59839 (2014)

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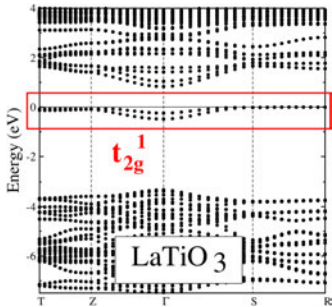
$t > 0.71$

Different structures ($R_A \sim R_B$)

Ilmenite

FeTiO3

Basics: TM electron shell



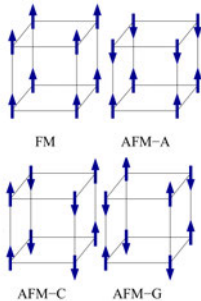
- Electrons: partially filled (3) d states

Electron localization (narrow bands, W)

Strong Coulomb correlations (U)

$U/W \rightarrow$ Mott–Hubbard insulator

Basics: TM electron shell



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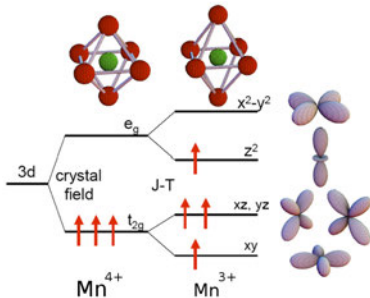
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Basics: TM electron shell



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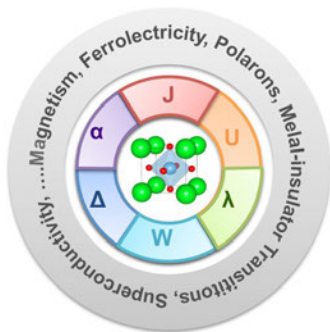
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Different types of magnetic orderings

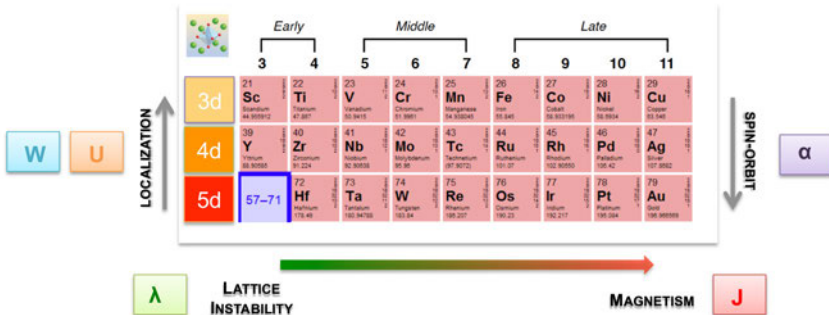
Jahn-Teller distortions in e_g band perovskites
(electron-phonon interaction)

Basics: Many (tunable) competing interactions



W	Bandwidth
Δ	Crystal Field
α	Spin-orbit
J	Hund's coupling
U	e-e repulsion
λ	e-ph interaction

Basics: Many (tunable) competing interactions



Competing and tunable energy scales give rise to a variety of functionalities

Basics: Applications

Wide spectrum of attractive properties $\Rightarrow$ large variety of functional behavior...

Property	Application	Material
Proton conductivity	SOFC electrolyte	BaCeO_3 , SrCeO_3 BaZrO_3
	Hydrogen sensor	
	H_2 production/extraction	
Ionic conductivity	Solid electrolyte	$(\text{La,Sr})(\text{Ga,Mg})\text{O}_{3-\delta}$
Mixed conductivity	SOFC electrode	$\text{La}(\text{Sr,Ca})\text{MnO}_{3-\delta}$, LaCoO_3 $(\text{La,Sr})(\text{Co,Fe})\text{O}_{3-\delta}$
Ferroelectric/ piezoelectric	Piezoelectric transducer	BaTiO_3 , $\text{Pb}(\text{Zr,Ti})\text{O}_3$, $\text{Pb}(\text{Mg,Nb})\text{O}_3$
	Thermistor, actuator	

Catalytic	Catalyst	LaFeO_3 , $\text{La}(\text{Ce,Co})\text{O}_3$
Electrical/ dielectric	Multilayer capacitor	BaTiO_3 , BaZrO_3
	Dielectric resonator	
	Thin film resistor	
Magnetic	Magnetic memory	GdFeO_3 , LaMnO_3
	Ferromagnetism	
Optical	Electrooptical modulator	$(\text{Pb,L a})(\text{Zr,Ti})\text{O}_3$
	Laser	YAlO_3 , KNbO_3
Super-conductivity	Superconductor	$\text{Ba}(\text{Pb,Bi})\text{O}_3$, BaKBiO_3

Basics: Applications

...and new emerging phenomena

- Oxide electronics: Two dimensional electron gas @ perovskite interface ($\text{SrTiO}_3 \parallel \text{LaAlO}_3$) and surface (SrTiO_3)
- Spintronics: half-metallic double perovskites
- Photovoltaics: Perovskite solar cells (Halide perovskites, $\text{CH}_3\text{NH}_3\text{PbX}_3$)

Computational Modeling

Realistic modeling of perovskites very challenging: the state of each electron **strongly** depends on the state of the other electrons of the system, which are coupled (or **correlate** with each other) via the Coulomb interaction.

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At the core of the problem there is the solution of the many body Hamiltonian.

Basic Hamiltonian of a system of N electrons and M nuclei

$$\begin{aligned}\hat{H} = & -\frac{\hbar^2}{2m_e} \sum_{i=1}^N \nabla_i^2 - \sum_{n=1}^M \frac{\hbar^2}{2M_n} \nabla_n^2 + \frac{1}{4\pi\epsilon_0} \frac{1}{2} \sum_{i,j=1; i \neq j}^N \frac{e^2}{|\mathbf{r}_i - \mathbf{r}_j|} \\ & - \frac{1}{4\pi\epsilon_0} \sum_n^M \sum_i^N \frac{Z_n e^2}{|\mathbf{r}_i - \mathbf{R}_n|} + \frac{1}{4\pi\epsilon_0} \frac{1}{2} \sum_{n,m=1; n \neq m}^M \frac{Z_n Z_m e^2}{|\mathbf{R}_n - \mathbf{R}_m|}\end{aligned}$$

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with

$$\hat{H} = \hat{T}_e$$

The kinetic energy operators for each electron in the system

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with

$$\hat{H} = \hat{T}_e + \hat{T}_n$$

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$$\hat{H} = \hat{T}_e + \hat{T}_n + \hat{U}_{ee}$$

The potential energy arising from Coulombic electron-electron repulsions

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The potential energy between the electrons and nuclei - the total electron-nucleus Coulombic attraction in the system

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The potential energy arising from Coulombic nuclei-nuclei repulsions

One Hamiltonian two communities

First principles (*Ab initio*)

Solution of the many body
Schrödinger equation

$$\hat{H}\Psi = E\Psi$$

- No empirical assumptions
- No fitting parameters*
- Full electronic structure**
- Different level of accuracy (approximations)
- Atomistic interpretation

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Model Hamiltonian

Solution of simplified lattice fermion models
(typically the Hubbard model)

$$H = -t \sum_{\langle i,j \rangle, \sigma} (c_{i,\sigma}^\dagger c_{j,\sigma} + c_{j,\sigma}^\dagger c_{i,\sigma}) + U \sum_{i=1}^N n_{i\uparrow} n_{i\downarrow}$$

- Restricted Hilbert space (few bands near E_F)
- Short-ranged electron interactions
- Adjustable parameters
- Accurate solution, transparent physical interpretation

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$$\text{FP+MH} \rightarrow \text{DFT(GW)} + \text{DMFT}$$

Approximations

$$\hat{H}\Psi = E\Psi$$

- ① Born-Oppenheimer approximation

$$\hat{H}_e\Psi = E\Psi, \quad \hat{H}_e = T_e + V_{ee} + V_{en}$$

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Many-body $\Psi(\mathbf{r}_1, \dots, \mathbf{r}_N)$ storage
requirements prohibitive

$$(\#\text{grid points})^N$$

② Map to "one-electron" theory

Density Functional Theory (LDA, GGA)

&

Hartree-Fock (EXX)

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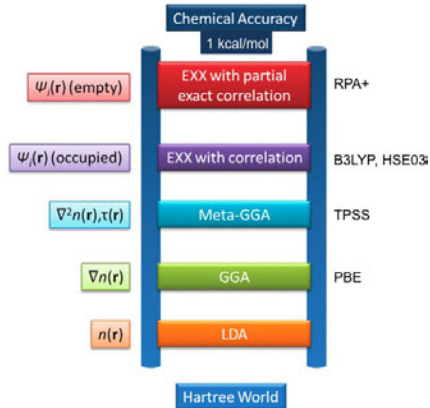
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The Jacob ladder



Solution of the full electronic Schrödinger equation

$$\hat{H}_e \Psi = E \Psi$$

Hartree: non-interacting (NI) particles (independent particle approximation)

$$\Psi^{HP}(\mathbf{x}_1, \mathbf{x}_2, \dots, \mathbf{x}_N) = \phi_1(\mathbf{x}_1) \phi_2(\mathbf{x}_2) \dots \phi_N(\mathbf{x}_N)$$

where $\phi(\mathbf{x})$ is an eigenstate of:

$$\hat{H}_e^{NI} = -\frac{\hbar^2}{2m_e} \sum_{i=1}^N \nabla_i^2 - \frac{1}{4\pi\epsilon_0} \sum_n^M \sum_i^N \frac{Z_n e^2}{|\mathbf{r}_i - \mathbf{R}_n|} = \sum_{i=1}^N h(\mathbf{r}_i)$$

Variational principle:

$$\langle H_e \rangle \equiv \frac{\langle \Psi^{HP} | \hat{H}_e | \Psi^{HP} \rangle}{\langle \Psi^{HP} | \Psi^{HP} \rangle} \geq E_0$$

The Hartree World:

$$\Psi^{HP}(\mathbf{x}_1, \mathbf{x}_2, \dots, \mathbf{x}_N) = \phi_1(\mathbf{x}_1) \phi_2(\mathbf{x}_2) \dots \phi_N(\mathbf{x}_N)$$

Limitation of the Hartree product:

① *Uncorrelated* probability distribution:

$$\rho(\mathbf{x}_1, \mathbf{x}_2, \dots, \mathbf{x}_N) = |\phi_1(\mathbf{x}_1)|^2 |\phi_2(\mathbf{x}_2)|^2 \dots |\phi_N(\mathbf{x}_N)|^2$$

The interaction among particles taken into account through the **Hartree term**,
i.e. electrostatic potential arising from the charge distribution of N electrons:

$$\sum_{i=1}^N \int dx' |\phi_i(\mathbf{x}')|^2 \frac{1}{|\mathbf{r} - \mathbf{r}'|}$$

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② *Self interaction error:*

The Hartree-term includes a sum over all electrons

③ Ψ^{HP} is *not Antisymmetric*

$$\mathcal{P}_{1,2} [\psi_1(\mathbf{x}_1) \psi_2(\mathbf{x}_2) \dots \psi_N(\mathbf{x}_N)] = \psi_1(\mathbf{x}_2) \psi_2(\mathbf{x}_1) \dots \psi_N(\mathbf{x}_N) \neq -\psi_1(\mathbf{x}_1) \psi_2(\mathbf{x}_2) \dots \psi_N(\mathbf{x}_N)$$

The Hartree-Fock Theory

In the HF method the WF the WF is written as a Slater determinant.

$$\begin{aligned} \Psi^{HP}(\mathbf{x}_1, \mathbf{x}_2, \dots, \mathbf{x}_N) &= \phi_1(\mathbf{x}_1) \phi_2(\mathbf{x}_2) \dots \phi_N(\mathbf{x}_N) \\ &\Downarrow \\ \Psi_{AS}(\mathbf{x}_1, \mathbf{x}_2, \dots, \mathbf{x}_N) &= \frac{1}{\sqrt{N!}} \begin{vmatrix} \phi_1(\mathbf{x}_1) & \phi_2(\mathbf{x}_1) & \dots & \phi_N(\mathbf{x}_1) \\ \phi_1(\mathbf{x}_2) & \phi_2(\mathbf{x}_2) & \dots & \phi_N(\mathbf{x}_2) \\ \vdots & \vdots & & \vdots \\ \phi_1(\mathbf{x}_N) & \phi_2(\mathbf{x}_N) & \dots & \phi_N(\mathbf{x}_N) \end{vmatrix} \end{aligned}$$

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- 1 **Antisymmetric**
- 2 **Self interaction free**
- 3 **Uncorrelated**

Only particles with the same spin have a certain degree of correlation:

$$\rho(\mathbf{x}_1, \mathbf{x}_2) = 0 \quad \text{if} \quad \mathbf{r}_1 = \mathbf{r}_2$$

Pauli principle

The Hartree-Fock Theory

The Hartree-Fock equations:

- ① Electronic Hamiltonian $\hat{H}_e = T_e + V_{ee} + V_{en}$

$$\hat{H}_e = -\frac{1}{2} \sum_{i=1}^N \nabla_i^2 + \frac{1}{2} \sum_{i,j=1; i \neq j}^N \frac{e^2}{|\mathbf{r}_i - \mathbf{r}_j|} - \sum_n^M \sum_i^N \frac{Z_n e^2}{|\mathbf{r}_i - \mathbf{R}_n|}$$

- ② WF $\rightarrow$ Slater determinant Ψ_{AS}

③ $\langle \Psi_{AS} | \hat{H}_e | \Psi_{AS} \rangle$

- ④ Variation principle



$$\left[-\frac{1}{2} \nabla^2 - \sum_n \frac{Z_n}{|\mathbf{r} - \mathbf{R}_n|} \right] \psi_k(\mathbf{x}) + \sum_{l=1}^N \int dx' |\psi_l(\mathbf{x}')|^2 \frac{1}{|\mathbf{r} - \mathbf{r}'|} \psi_k(\mathbf{x}) \\ - \sum_{l=1}^N \int dx' \psi_l^*(\mathbf{x}') \frac{1}{|\mathbf{r} - \mathbf{r}'|} \psi_k(\mathbf{x}') \psi_l(\mathbf{x}) = \epsilon_k \psi_k$$

It has the form of a Schrödinger equation, where the eigenvalues ϵ_k represent the orbital energies. One more term: **EXCHANGE**.

The Hartree-Fock Theory

$$\hat{\mathcal{F}}\psi_k = \left[-\frac{1}{2}\nabla^2 - \sum_n \frac{Z_n}{|\mathbf{r} - \mathbf{R}_n|} \right] \psi_k(\mathbf{x}) + \sum_{l=1}^N \int dx' |\psi_l(\mathbf{x}')|^2 \frac{1}{|\mathbf{r} - \mathbf{r}'|} \psi_k(\mathbf{x}) \\ - \sum_{l=1}^N \int dx' \psi_l^*(\mathbf{x}') \frac{1}{|\mathbf{r} - \mathbf{r}'|} \psi_k(\mathbf{x}') \psi_l(\mathbf{x}) = \epsilon_k \psi_k$$

- ① **Hartree term only**: unphysical coupling between orbital k and itself (self-energy)
- ② **Exchange term non-local**: its value at $\mathbf{r}$ depends on $\psi_k(\mathbf{r}')$

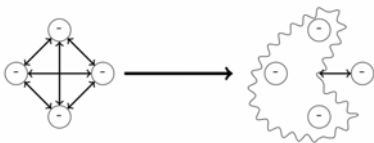
The **Exchange term** is the same as the **Hartree term**, with two spin-orbitals labels k and l interchanged and a *minus* sign.

- ③ **Fock+Hartree**: self-energy problem solved

The Fock operator $\hat{\mathcal{F}}$ is a one particle operator. It replaces the many-body Schrödinger equation by a set of one-particle equations, in which each electron moves in an effective field, often also called *mean-field*.

The Hartree-Fock Theory

The Fock operator $\hat{\mathcal{F}}$ is a one particle operator. It replaces the many-body Schrödinger equation by a set of one-particle equations, in which each electron moves in an effective *mean-field*.



Many-body problem $\rightarrow$ one-electron problems

The WF does not have explicit dependence on the position of all other electrons. The dependence is only implicit via the Exchange term.

Density Functional Theory

Hartree-Fock equations:

$$\left(-\frac{\hbar^2}{2m_e} \nabla^2 + V_e(\mathbf{r}) + V_H(\mathbf{r}) \right) \phi_i(\mathbf{r}) + \int V_x(\mathbf{r}, \mathbf{r}') \phi_i(\mathbf{r}') d^3 r' = \epsilon_i^{HF} \phi_i(\mathbf{r})$$

- Mean-field
- WF-based method: Slater determinant
- *exact* exchange V_x
- no correlation (only Pauli principle)
- Self-interaction free

Density Functional Theory

Hartree-Fock equations:

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DFT equations:

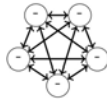
$$\left(-\frac{\hbar^2}{2m_e} \nabla^2 + V_{en}(\mathbf{r}) + V_H + V_{xc}(\mathbf{r}) \right) \phi_i(\mathbf{r}) = \epsilon_i^{KS} \phi_i(\mathbf{r})$$

- 1 Mean-field
- 2 Density-based method
- 3 *approximate* exchange & correlation, V_{xc}
- 4 Self-interaction problem

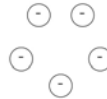
Density Functional Theory

1 Mean-field

The real interacting system is replaced by an effective Kohn-Sham (KS) system of non-interacting electrons in an effective potential V .



$\mathcal{H}$



$$h_{\text{aux}} = -\frac{\nabla^2}{2} + V(\mathbf{x})$$

$$H \rightarrow \sum_{i=1}^N h_{\text{aux}}; \quad h_{\text{aux}}\phi_i = \epsilon_i\phi_i; \quad \phi_i = \text{KS single-particle orbitals}$$

2 Density-based method

The DFT approach promotes the particle density

$$n(\mathbf{r}) = \sum_{i, \text{occupied}} |\phi_i(\mathbf{r})|^2.$$

to the status of key variable, on which the calculation of all other observables can be based (Hohenberg-Kohn Theorem)

$$\begin{array}{ccc} V_{\text{ext}}(\mathbf{r}) & \xleftarrow{HK} & n_0(\mathbf{r}) \\ \Downarrow & & \Uparrow \\ \Psi_i(\mathbf{r}) & \Rightarrow & \Psi_0(\mathbf{r}) \end{array}$$

Density Functional Theory

③ *approximate* exchange and correlation, V_{xc}

Universal functional for the energy given in terms of the density:

$$E[n] = T_{KS}[n] + \int d^3r n(\mathbf{r}) V_{ext}(\mathbf{r}) + \frac{e^2}{2} \iint d^3r d^3r' \frac{n(\mathbf{r})n(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} + E_{xc}[n]$$

$$E[n] \geq E[n_{GS}] = E_{GS}^{True}$$

The quality of the solution depend on the (*unknown*) form of the XC potential V_{xc} :

$$V_{xc}([n], \mathbf{r}) = \frac{\delta E_{XC}}{\delta n(\mathbf{r})}$$

Approximations needed:

$$E_{xc}^{LDA}[n] = \int d^3r e_{xc}(n(\mathbf{r}))$$

$$E_{xc}^{GGA}[n] = \int d^3r f(n(\mathbf{r}), \nabla n(\mathbf{r}))$$

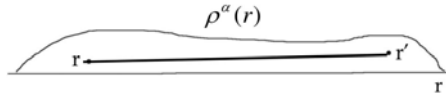
Density Functional Theory

4 Self-interaction problem

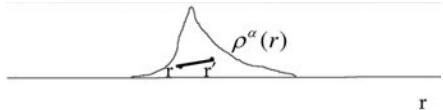
Non complete cancellation of the spurious repulsion of each electron from itself, included in the Hartree term, that is not completely accounted for in XC functionals:

$$\Delta\epsilon = \iint \frac{n_1(\mathbf{r})n_1(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} d\mathbf{r}d\mathbf{r}'$$

Delocalized States $\rightarrow$ small Errors



Localized States $\rightarrow$ large Errors



Hands-on: DFT & HF (VASP)

Hartree-Fock equations:

$$\left(-\frac{\hbar^2}{2m_e} \nabla^2 + V_e(\mathbf{r}) + V_H(\mathbf{r}) \right) \phi_i(\mathbf{r}) + \int V_x(\mathbf{r}, \mathbf{r}') \phi_i(\mathbf{r}') d^3r' = \epsilon_i^{HF} \phi_i(\mathbf{r})$$

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- WF-based method: Slater determinant
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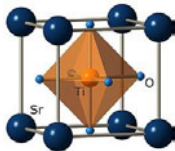
DFT equations:

$$\left(-\frac{\hbar^2}{2m_e} \nabla^2 + V_{en}(\mathbf{r}) + V_H + V_{xc}(\mathbf{r}) \right) \phi_i(\mathbf{r}) = \epsilon_i^{KS} \phi_i(\mathbf{r})$$

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- 2 Density-based method
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- 4 Self-interaction problem

Hands-on: DFT & HF (VASP)

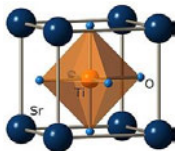
Test case: SrTiO_3



	DFT(GGA)	HF	Expt.
Direct gap (eV)	?	?	3.75
$\Gamma - \Gamma$			
Running time	?	?	

Hands-on: DFT & HF (VASP)

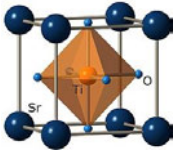
Test case: SrTiO_3



	DFT(GGA)	HF	Expt.
Direct gap (eV) ($\Gamma - \Gamma$)	2.16	4.66	3.75
CPU time (s)	5	613	

Hands-on: DFT & HF (VASP)

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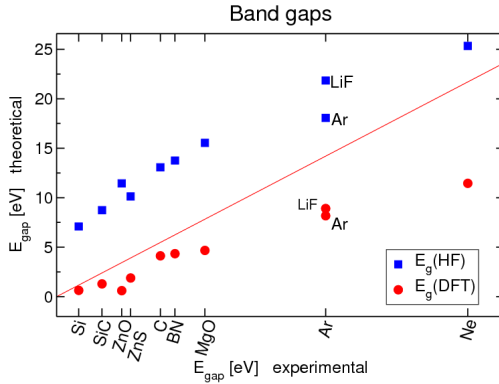
DFT

- 1 band gap too small
(self-interaction problem)
- 2 fast
(approx. E_{xc})

HF

- 1 band gap too large
(no correlation)
- 2 slow
(two-particle integral, exact exchange)

DFT vs. HF

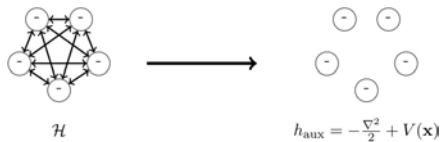


General trend: DFT tends to underestimate band gaps, HF overestimates

DFT + HF ?

Hybrid Functionals

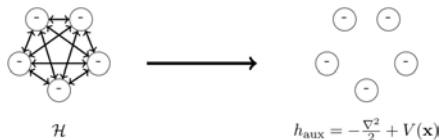
Adiabatic connection and coupling-constant integration



The adiabatic connection continuously transform the real interacting system to the non-interacting one

Hybrid Functionals

Adiabatic connection and coupling-constant integration



The adiabatic connection continuously transform the real interacting system to the non-interacting one

$$V_{ee} \rightarrow \lambda V_{ee} \quad \text{with} \quad 0 \leq \lambda \leq 1$$

$\lambda = 0$: non-interacting KS system $\rightarrow \Psi_{\lambda=0} = \text{KS-WF}$

$\lambda = 1$: exact many-body systems $\rightarrow \Psi_{\lambda=1} = \text{Exact many-body WF}$

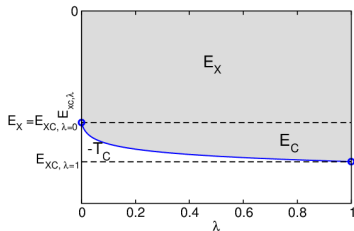
$$E_{XC} = \int_0^1 E_{xc,\lambda} d\lambda, \quad E_{xc,\lambda} = \langle \Psi_\lambda | V_{ee} | \Psi_\lambda \rangle - E_H$$

Hybrid Functionals

Adiabatic connection and coupling-constant integration

$$E_{XC} = \int_0^1 E_{xc,\lambda} d\lambda, \quad E_{xc,\lambda} = \langle \Psi_\lambda | V_{ee} | \Psi_\lambda \rangle - E_H$$

$E_{xc,\lambda}$ is a monotonically decreasing function of λ



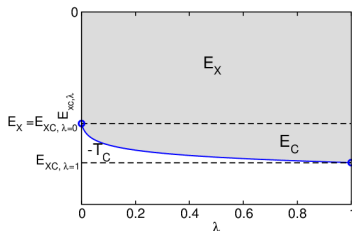
The area between the curve and the x-axis is just E_{xc} .

Hybrid Functionals

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The integral can be approximated via linear interpolation:

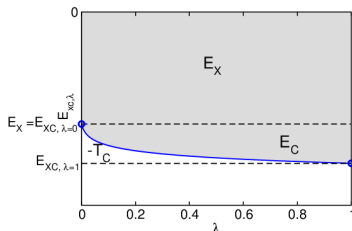
$$E_{XC} \approx \frac{1}{2} E_{XC, \lambda=0} + \frac{1}{2} E_{XC, \lambda=1}$$

Hybrid Functionals

Adiabatic connection and coupling-constant integration

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$$E_{XC} \approx \frac{1}{2} E_{XC,\lambda=0} + \frac{1}{2} E_{XC,\lambda=1} \Rightarrow E_{XC}^{\text{Hybrid}} = \frac{1}{2} E_X^{\text{HF}} + \frac{1}{2} E_{XC}^{\text{LDA}}$$

The Hybrid Functionals Family

Half-Half, Becke 1993

$$E_{XC}^{\text{Hybrid}} = \frac{1}{2}E_X^{HF} + \frac{1}{2}E_{XC}^{LDA}$$

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B3LYP, Becke 1993-1994

$$E_{XC}^{\text{B3LYP}} = E_X^{LDA} + \alpha_1(E_X^{HF} - E_X^{LDA}) + \alpha_2(E_X^{GGA} - E_X^{LDA}) + \alpha_3(E_C^{GGA} - E_C^{LDA})$$

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PBE0, Perdew-Burke-Ernzerhof 1996

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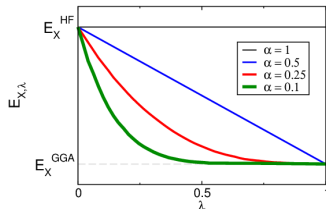
Choice of α

Optimum: $\alpha = 0.25$ (fit to MP2) $\rightarrow$

In practice α is system dependent

Disadvantages

Long-range E_X numerically complicated



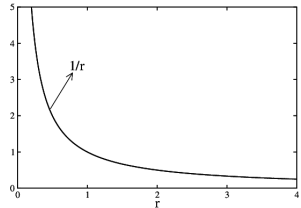
Screened Hybrid Functionals: HSE

From PBE0 to HSE

$$V_x \rightarrow V_{sx}$$

$$-e^2 \frac{\sum_j f_j \phi_j(\mathbf{r}) \phi_j^*(\mathbf{r}')}{|\mathbf{r} - \mathbf{r}'|} \rightarrow -e^2 \sum_j f_j \phi_j(\mathbf{r}) \phi_j^*(\mathbf{r}') \frac{\text{erfc}(\mu |\mathbf{r} - \mathbf{r}'|)}{|\mathbf{r} - \mathbf{r}'|}$$

$$\frac{1}{r} = \underbrace{\frac{\text{erfc}(\mu r)}{r}}_{sr} + \underbrace{\frac{\text{erf}(\mu r)}{r}}_{lr}$$



$$E_{XC}^{\text{HSE}} = \alpha E_X^{\text{HF},sr}(\mu) + (1 - \alpha) E_X^{\text{PBE},sr}(\mu) + \alpha E_X^{\text{PBE},lr}(\mu) + E_C^{\text{PBE}}$$

Default values:

$$\alpha = 0.25,$$

$$\mu = 0.2 \text{ \AA}^{-1}$$

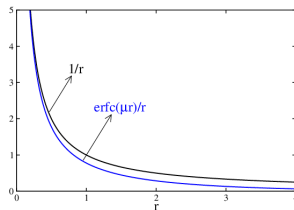
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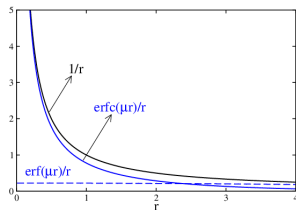
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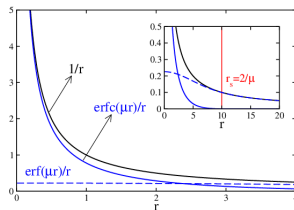
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Screened Hybrid Functionals

Why Screening

- Computational convenience: neglect numerically complex long-range part of E_X
- Physical justification: in many-body systems the unscreened exchange is reduced

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Hybrid Functionals meets GW

- DFT

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{xc}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) = \epsilon_{n\mathbf{k}} \psi_{n\mathbf{k}}(\mathbf{r})$$

- HF

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int V_{\text{X}}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' = \epsilon_{n\mathbf{k}} \psi_{n\mathbf{k}}(\mathbf{r})$$

- HSE

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{c}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int V_{\text{sX}}^{\text{HSE}}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' = \epsilon_{n\mathbf{k}} \psi_{n\mathbf{k}}(\mathbf{r})$$

- Quasiparticle equations (GW)

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma(\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}}) \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' = \epsilon_{n\mathbf{k}} \psi_{n\mathbf{k}}(\mathbf{r})$$

Screened Hybrid Functionals

Hybrid Functionals meets GW

- HSE

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{c}}(\mathbf{r}) + \alpha V_{\text{x}}^{lr}(\mathbf{r}) + (1 - \alpha) V_{\text{x}}^{sr}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \alpha V_{\text{x}}^{sr}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}'$$

- Quasiparticle equations (GW)

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma(\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}}) \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}'$$

$$\Sigma(\omega) = iG(\omega)W(\omega)$$

W = screened (GW) Coulomb interaction

$$W(\omega) = \epsilon(\omega)^{-1}v \xrightarrow{\epsilon(\omega) \rightarrow \epsilon_{\infty} = 1/\alpha} \alpha v \xrightarrow{v \rightarrow v_{sr}} \alpha \frac{\text{erfc}(\mu|\mathbf{r} - \mathbf{r}'|)}{|\mathbf{r} - \mathbf{r}'|}$$

Screened Hybrid Functionals

Hybrid Functionals meets GW

- HSE

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{c}}(\mathbf{r}) + \alpha V_{\text{x}}^{lr}(\mathbf{r}) + (1 - \alpha)V_{\text{x}}^{sr}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \alpha V_{\text{x}}^{sr}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}'$$

- Quasiparticle equations (GW)

$$\left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r})\right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma(\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}}) \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}'$$

$$\Sigma(\omega) = iG(\omega)W(\omega)$$

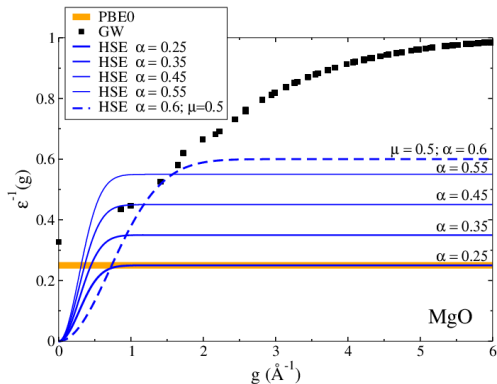
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$$W(\omega) = \epsilon(\omega)^{-1}v \xrightarrow{\epsilon(\omega) \rightarrow \epsilon_{\infty} = 1/\alpha} \alpha v \xrightarrow{v \rightarrow v_{sr}} \alpha \frac{\text{erfc}(\mu|\mathbf{r} - \mathbf{r}'|)}{|\mathbf{r} - \mathbf{r}'|}$$

$$\text{Optimum } \alpha = \frac{1}{\epsilon_{\infty}}$$

Hybrid Functionals

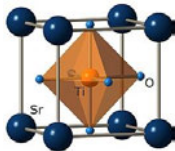
but HSE is NOT GW!



- No Exchange at short wave vector (good only for metals)
- 0.25 choice questionable for wide band gap (small ϵ_∞) materials
better choice $\alpha = \frac{1}{\epsilon_\infty}$
- α and μ fitting parameters

Hands-on: HSE (VASP)

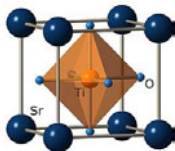
Test case: SrTiO₃



	DFT(GGA)	HF	HSE	Expt.
Direct gap (eV) ($\Gamma - \Gamma$)	2.16	4.66	?	3.75
CPU time (s)	5	613	?	

Hands-on: HSE (VASP)

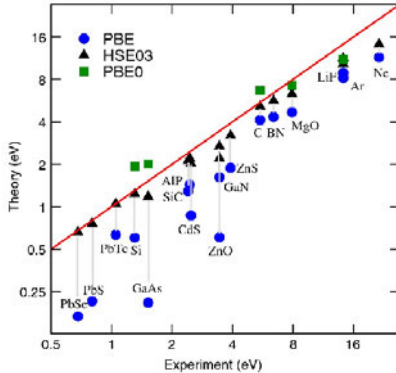
Test case: SrTiO₃



	DFT(GGA)	HF	HSE	Expt.
Direct gap (eV) ($\Gamma - \Gamma$)	2.16	4.66	3.8	3.75
CPU time (s)	5	613	324	

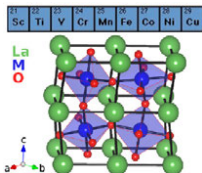
DFT vs. PBE0/HSE

Semiconductors dataset ($\alpha = 0.25$)

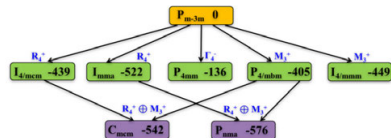
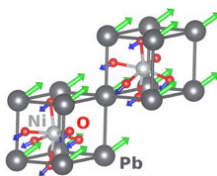
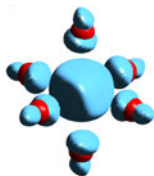
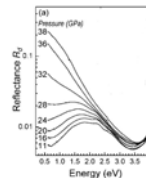


Results

Screened Hybrid Functionals applied to Perovskites

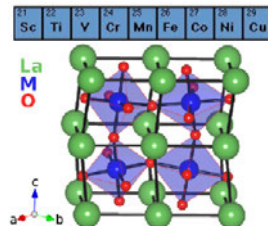


25 Mn Manganese 54.938049	26 Fe Iron 55.845	27 Co Cobalt 58.933200
43 Tc Technetium (98)	44 Ru Ruthenium 101.07	45 Rh Rhodium 102.90550
75 Re Rhenium 186.207	76 Os Osmium 190.23	77 Ir Iridium 192.217



Results: the 3d LaMO_3 perovskite dataset

- Band, Mott-Hubbard, charge-transfer insulator
- NM, A/C/G-AFM, PM
- Orthorhombic, Monoclinic, Rhombohedral, Tetragonal
- t_{2g} , e_g : different degree of (de)localization



LaScO_3	LaTiO_3	LaVO_3	LaCrO_3	LaMnO_3	LaFeO_3	LaCoO_3	LaNiO_3	LaCuO_3
O d^0 	O ↑ 	M ↑↑ 	O ↑↑↑ 	O ↑↑↑ ↑ 	O ↑↑↑ ↑↑ 	R ↑↓↑↓↑↓ 	R ↑↓↑↓↑↓ ↑ M	T ↑↓↑↓↑↓ ↑↓ M
NM	G-AFM	C-AFM	G-AFM	A-AFM	G-AFM	PM	PM	PM

LDA+U: I. Solovyev, N. Hamada, and K. Terakura, Phys. Rev. B **53**, 7158 (1996).

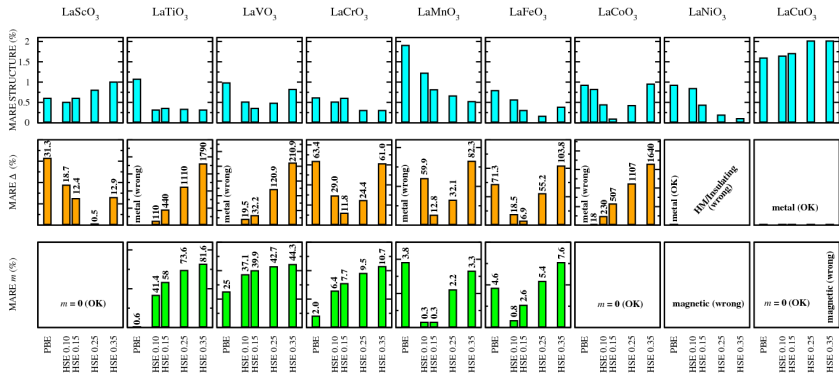
HF: T. Mizokawa and A. Fujimori, Phys. Rev. **54**, 5368 (1996).

GW: Y. Nohara, S. Yamamoto, and T. Fujiwara, Phys. Rev. B **79**, 195110 (2009).

HSE: J. He & C. Franchini, Phys. Rev. B **86**, 235117 (2012).

Results: the 3d LaMO_3 perovskite dataset

Mean absolute relative value (MARE) based on a fitting of **structural properties** (volume, lattice parameters, JT/GFO, ion positions), **band gap**, **magnetic moment**, **spin ordering**.



Fitting α

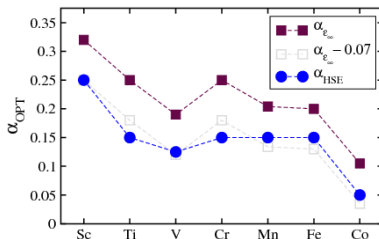
	LaScO_3	LaTiO_3	LaVO_3	LaCrO_3	LaMnO_3	LaFeO_3	LaCoO_3	LaNiO_3	LaCuO_3
α	0.25	0.10 (0.15)	0.10-0.15	0.15	0.15	0.15	0.05	0.0	0.0

Results: the 3d LaMO₃ perovskite dataset

Fitting vs $\alpha_{opt} = 1/\epsilon_{\infty}$

① $\alpha_{opt}^{HSE} \rightarrow$ HSE fitting

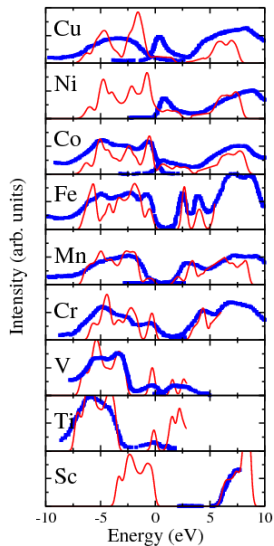
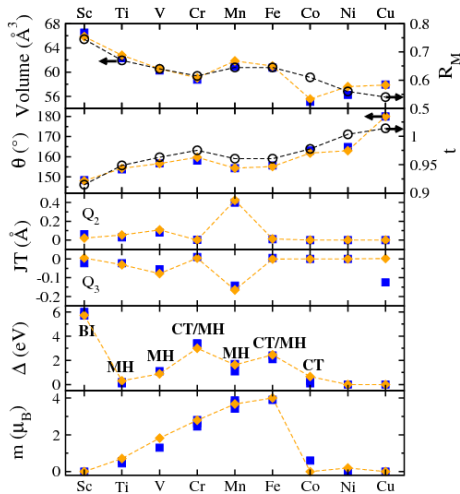
② $\alpha_{opt}^{\epsilon_{\infty}} = \frac{1}{\epsilon_{\infty}}$



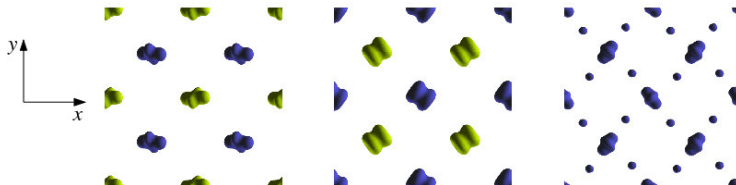
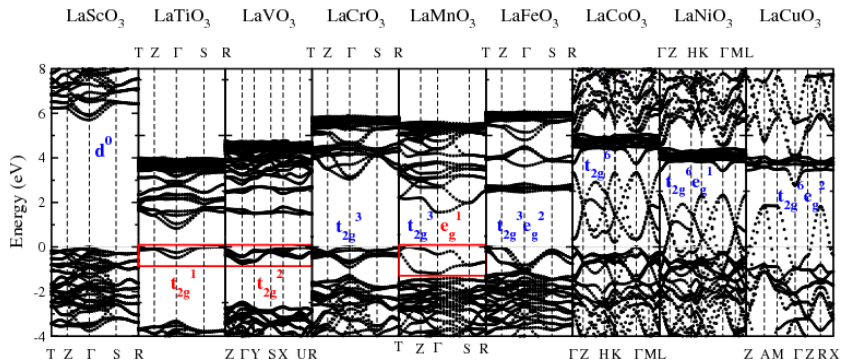
	LSO	LTO	LVO	LCrO	LMO ₃	LFO	LCoO	LNO	LCuO
α_{opt}^{HSE}	0.25	0.10/0.15	0.10-0.15	0.15	0.15	0.15	0.05	0	0
$\alpha_{opt}^{\epsilon_{\infty}}$	0.323	0.125	0.192	0.250	0.204	0.200	0.105	0	0
(ϵ_{∞})	(3.1)	(8.0)	(5.2)	(4.0)	(4.9)	(5.0)	(9.5)	∞	∞

0.07 shift $\rightarrow$ HSE (screened) already contains some screening (μ)

Results: the 3d LaMO_3 perovskite dataset



Results: the 3d LaMO_3 perovskite dataset



Results: $4d$ and $5d$ perovskites

Larger spatial extension $\Rightarrow$ Reduced correlation, Metallic & "itinerant magnetism"

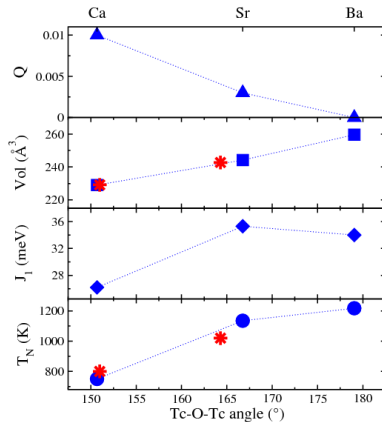
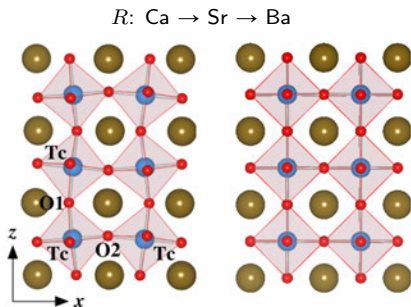
But...

25 Mn Manganese 54.938049	26 Fe Iron 55.845	27 Co Cobalt 58.933200
43 Tc Technetium (98)	44 Ru Ruthenium 101.07	45 Rh Rhodium 102.90550
75 Re Rhenium 186.207	76 Os Osmium 190.23	77 Ir Iridium 192.217

- High T_N in CaTcO_3 and SrTcO_3
M. Avdeev *et al.*, JACS **133** 1654 (2011).
E.E. Rodriguez *et al.*, PRL **106** 067201 (2011).
- insulating state/weak magnetism in BaIrO_3
M.A. Laguna-Marco *et al.*, PRL **105** 216407 (2010).
R. Arita *et al.*, PRL **108**, 086403 (2012).
- Relativistic Mott state in Sr_2IrO_4
B.J. Kim *et al.*, PRL **101** 076402 (2008).
- Magnetic insulator NdOsO_3
Y. Shi *et al.* PRB **80**, 161104(R) (2009).

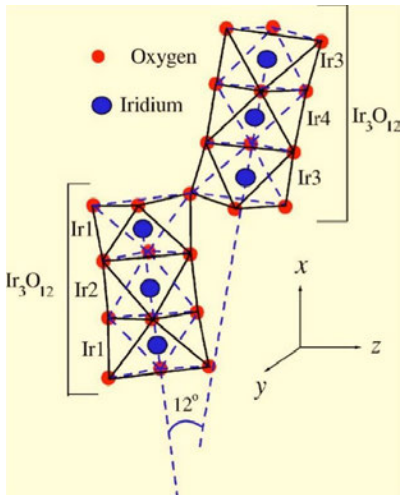
Results: 4d $RTcO_3$ perovskites

$$\text{HSE total energies} + -\sum_{i<j} \mathbf{J}_{ij} \mathbf{S}_i \cdot \mathbf{S}_j + \text{Monte Carlo}$$



BaTcO_3 : the highest T_N for compounds without 3d transition metals

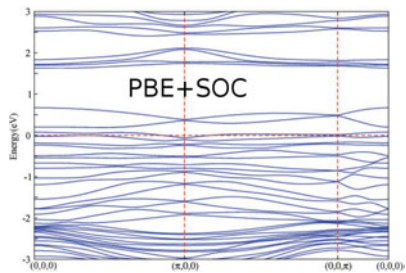
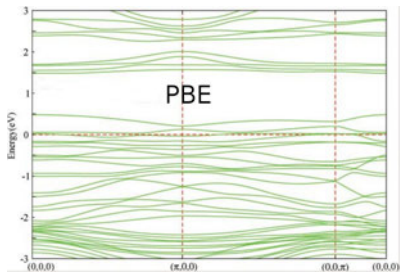
Results: $5d$ BaIrO_3 perovskite



- Quasi 1D
- Weakly ferromagnetic $\mu \approx 0.03\mu_B$
- Nonmetallic (gap $\approx 10\text{-}50$ meV)
- Strong spin-orbit interaction

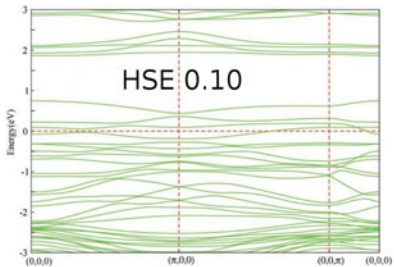
M. A. Laguna-Marco, *et al.*, PRL **105**, 216407 (2010).

Results: 5d BaIrO₃ perovskite

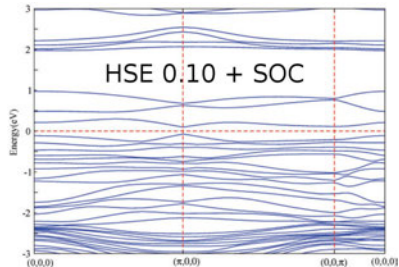


METAL !

Results: 5d BaIrO₃ perovskite

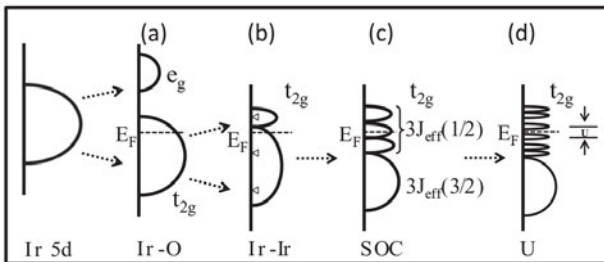


METAL !



INSULATOR, $\mu \approx 0.1\mu_B$

Results: 5d BaIrO_3 perovskite



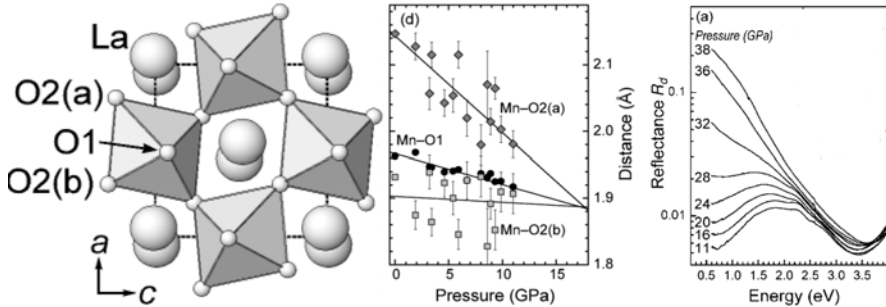
(a) Crystal field splitting: e_g , t_{2g}

(b) Ir-Ir interaction

(c) Spin-orbit coupling

(e) Moderate U

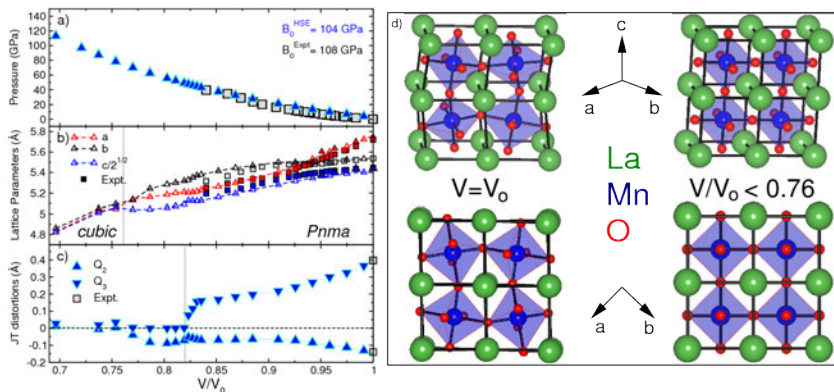
Results: MIT, LaMnO_3 under pressure



Loa *et al.* PRL **87**, 125501 (2001)

- 1 Progressive reduction of the JT distortions
- 2 Insulator to metal transition (Mott ?, i.e. JT distortions in the insulating phase?)

Results: MIT, LaMnO_3 under pressure



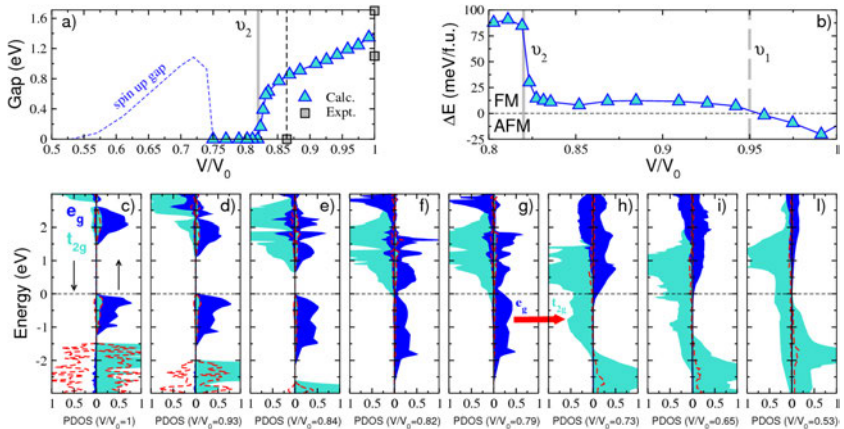
1 Excellent agreement with Experiment

Loa *et al.* PRL **87**, 125501 (2001); L. Pinsard-Gaudart *et al.* PRB **64**, 064426 (2001)

2 Progressive quenching of JT distortions

3 Orthorhombic to Cubic transition

Results: MIT, LaMnO_3 under pressure



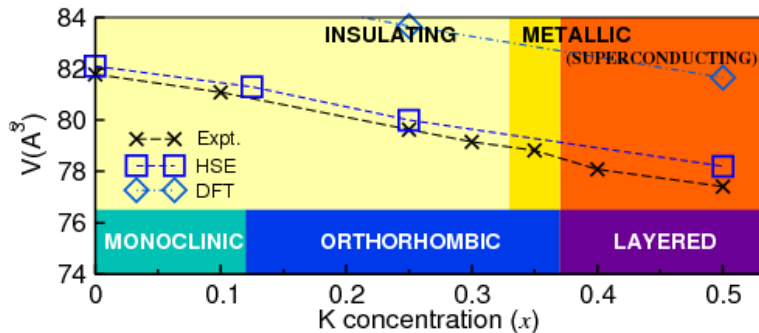
- 1 $V/V_0=0.82 \Rightarrow$ non-Mott Insulator to metal transition

Trimarchi & Binggeli PRB 2005 (LDA+U); Yamasaki *et al.* PRL 2006 (DMFT)

- 2 Intermediate pressure $\Rightarrow$ AFM (distorted) FM (undistorted) competition
- 3 $V/V_0=0.70 \Rightarrow$ AFM to FM (& Low-spin to High spin) transition

Results: polarons in BaBiO₃

hole-doping driven MIT



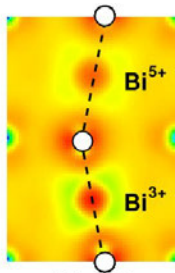
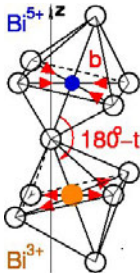
C. Franchini *et al.* PRL 102, 256402 (2009).

C. Franchini *et al.* PRB 81, 085213 (2010).

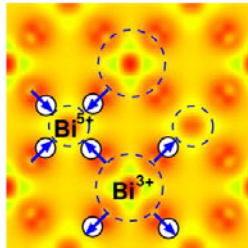
Results: polarons in BaBiO_3

$$x=0$$

- Valence skipping $\Rightarrow$ Charge disproportionation ($\text{Bi}^{4+} \Rightarrow \text{Bi}^{5+}/\text{Bi}^{3+}$, multivalency)
- Structural distortions $\Rightarrow$ breathing (b) and tilting (t) instabilities
- Charged ordered (CDW) insulating state



tilting (t)

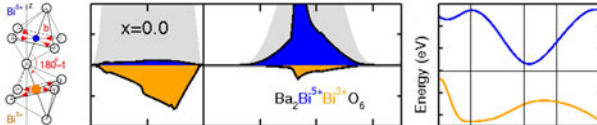


breathing (b)

Results: polarons in BaBiO₃

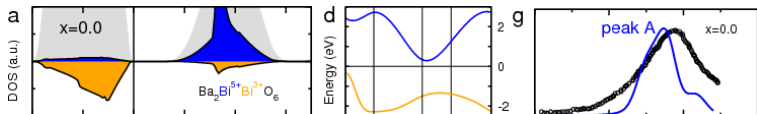
$$x=0$$

- Valence skipping $\Rightarrow$ Charge disproportionation
($\text{Bi}^{4+} \Rightarrow \text{Bi}^{5+}/\text{Bi}^{3+}$, multivalency)
- Structural distortions $\Rightarrow$ breathing (b) and tilting (t) instabilities
- Charged ordered (CDW) insulating state



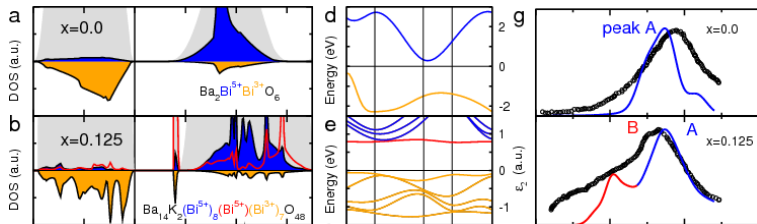
Results: polarons in BaBiO₃

$$x > 0$$



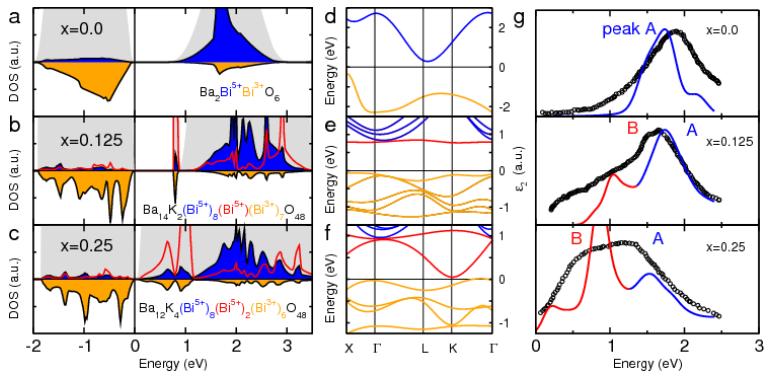
Results: polarons in BaBiO₃

$$x > 0$$



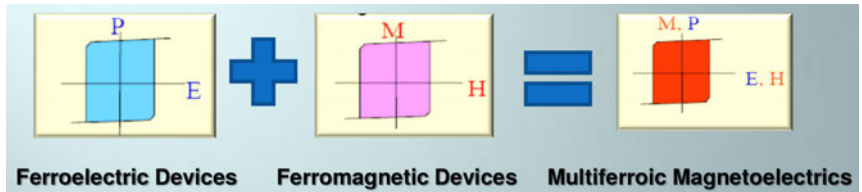
Results: polarons in BaBiO₃

$$x > 0$$



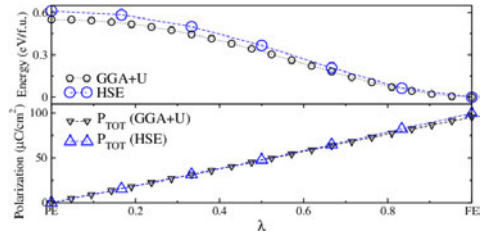
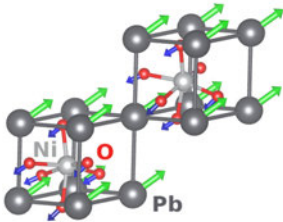
Results: Magnetism + Ferroelectricity

MULTIFERROICS



Results: Magnetism + Ferroelectricity

HSE + modern theory of polarization



LiNbO₃-type structure (*R3c*)

Arrows: FE displacements

Total polarization $P_{\text{tot}} \sim 100 \mu\text{C}/\text{cm}^2$

Larger than BiFeO₃

Results: Crystal Structure Prediction

HSE + lattice dynamics + Group Theoretical Methods

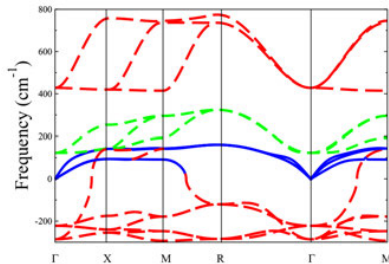
SrPdO_3 : New perovskite created in 2010 (Journal of Power Sources **195** 3806)

Results: Crystal Structure Prediction

HSE + lattice dynamics + Group Theoretical Methods

SrPdO₃: New perovskite created in 2010 (Journal of Power Sources **195** 3806)

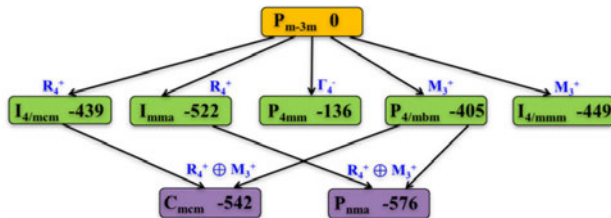
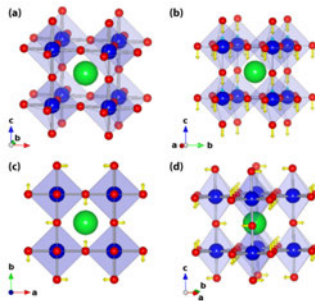
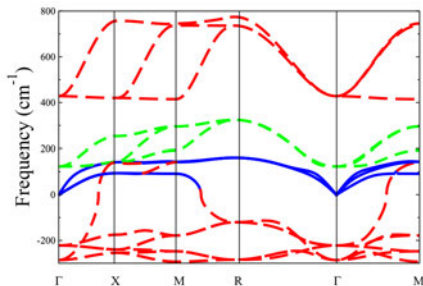
Structure: Unknown ($t=0.905 \rightarrow$ cubic phase unstable)



Strategy: stabilization of the negative modes (Γ_4^- , M_3^+ , R_4^+) through atomic displacements.

Results: Crystal Structure Prediction

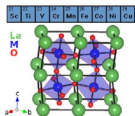
P_{nma}



$$+ \int \alpha V_{\mathbf{x}}^{sr}(\mathbf{I}$$

$$\begin{aligned}
& \left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{xc}}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) \\
& + \\
& \left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int V_{\text{X}}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' \\
& = \\
& \left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) + V_{\text{c}}(\mathbf{r}) + \alpha V_{\text{x}}^{lr}(\mathbf{r}) + (1 - \alpha) V_{\text{x}}^{sr}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \alpha V_{\text{x}}^{sr}(\mathbf{r}, \mathbf{r}') \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}' \\
& \approx \\
& \left(-\frac{1}{2}\Delta + V_{\text{ext}}(\mathbf{r}) + V_{\text{H}}(\mathbf{r}) \right) \psi_{n\mathbf{k}}(\mathbf{r}) + \int \Sigma(\mathbf{r}, \mathbf{r}', E_{n\mathbf{k}}) \psi_{n\mathbf{k}}(\mathbf{r}') d\mathbf{r}'
\end{aligned}$$

Hybrid functionals for materials science: accurate and predictive



25 Mn Manganese 54.938045	26 Fe Iron 55.845	27 Co Cobalt 58.933200
43 Tc Technetium (98)	44 Ru Ruthenium 101.07	45 Rh Rhodium 102.90550
75 Re Rhenium 186.207	76 Os Osmium 190.23	77 Ir Iridium 192.227

